

## Erratum: “Response to ‘Comment on ‘Deformation mechanisms of face-centered-cubic metal nanowires with twin boundaries’” [Appl. Phys. Lett. 93, 086102 (2008)]

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We have mistakenly stated that the strain of  $\varepsilon=0.37$ ,  $\varepsilon=0.43$ , and  $\varepsilon=0.47$  in the bottom of Fig. 1 and not indicated their meaning. We mean the yield strain and it should be  $\varepsilon_y=0.037$ ,  $\varepsilon_y=0.043$ , and  $\varepsilon_y=0.047$  for the three wires, correspondingly. To be clear, Fig. 1 should change and the caption appears correctly below.

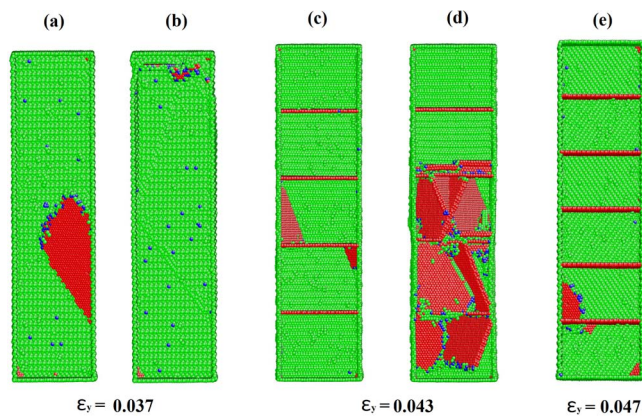


FIG. 1. (Color online) The snapshots corresponding to the first yield point of the stress-strain curves of the three wires. For the twin-free single-crystal Cu nanowire, (a) The nucleation of the first dislocation; (b) several picoseconds later after relaxation, dislocations move out of the wire, showing the dislocation starvation in the single-crystal wire. For the four-twin nanowire, (c) The nucleation of the first dislocations; (d) after relaxation a number of dislocations pile up at twin boundaries and stacking fault intersections, leading to some hardening effects after yielding. (e) The nucleation of first dislocation in the five-twin nanowire. The yield strain ( $\varepsilon_y$ ) for the twin-free wire, the four-twin wire as well as the five-twin is 0.037, 0.043, and 0.047, respectively. This clearly shows that the first dislocation emission in twinned nanowires, occur later than that of the twin-free one.